

Chemical composition and band bending at Al₂O₃/GaAs interface formed via *in situ* Al₂O₃ atomic layer deposition on pristine GaAs

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We report on the x-ray photoelectron spectroscopy (XPS) study of Al₂O₃/GaAs interface obtained by *in situ* atomic layer deposition (ALD) of Al₂O₃ on pristine (001) oriented GaAs. The *in situ* approach enables us to benefit from the contamination-free GaAs surface at the beginning of the Al₂O₃ deposition. We are going beyond the scopes of the previous studies [1] and examine the *in situ* Al₂O₃ ALD process in detail, namely varying sequence and number of precursors' pulses at the deposition beginning, deposition temperature, thickness of the deposited Al₂O₃ layer and also the surface termination of the GaAs. The Al₂O₃/GaAs structures (Fig. 1 (a), inset) are fabricated in a multi-chamber growth system, which includes a Scienta Omicron molecular-beam epitaxy (MBE) chamber and a FlexAL®II ALD system connected via transfer modules. MBE is employed to grow non-intentionally *p*-doped GaAs layer onto a GaAs (001) wafer. We deposit Al₂O₃ using trimethylaluminium (TMA) as a metal precursor, the oxygen source being deionized water. The characterization of the Al₂O₃/GaAs interface is performed *ex situ* by XPS with an Al K α (1486.6 eV) X-ray source.

High-resolution spectra are taken for arsenic (As 3*d* and As 2*p*_{3/2}), gallium (Ga 3*d* and Ga 2*p*_{3/2}), aluminium (Al 2*p*) and oxygen (O 1*s*) using a step size of 0.1 eV and pass energy of 23.5 eV. The inset of Fig. 1 (b) represents the Al 2*p* spectrum with only one peak assigned to Al-O bond in Al₂O₃ and this confirms the successful growth of Al₂O₃ on the GaAs surface. The O/Al ratio calculated from the measured Al 2*p* and O 1*s* spectra varies from 1.4 to 1.6 and indicates high stoichiometry of the grown oxide. We show that in the case of 'standard' Al₂O₃ deposition (Fig. 1 (a)) on the As-terminated GaAs surface, which begins with a TMA pulse, followed by a water pulse, the Al₂O₃/GaAs interface is free from high-valence arsenic (As₂O₃ and As₂O₅) and gallium (Ga₂O₃) oxides if the Al₂O₃ layer has a thickness of 5 nm and more. However, we reveal formation of low-valence arsenic oxides (As²⁺ and As¹⁺ oxidation states (peak As-Y in Fig. 1 (a))). The Al₂O₃ deposition performed on the Ga-terminated GaAs surface and started with 5H₂O pulses leads to the formation of a completely Ga/As-oxide-free Al₂O₃/GaAs interface (Fig. 1(b)). Analysis of the XPS data shows that the band bending in GaAs at the Al₂O₃/GaAs interface is reduced by 0.3 eV compared to the bending in the case of not passivated GaAs surface. This indicates the reduction of the interface states density, i.e. unpinned state of the Fermi level at the *in situ*-Al₂O₃/GaAs interface. Photoluminescence measurements of Al₂O₃/GaAs structures confirm the reduced density of the interface states. Our study delivers criteria for high-quality passivation of GaAs surface.

References

References

[1] T. W. Pi et al., Nanotechnology **26**, 164001 (2015).

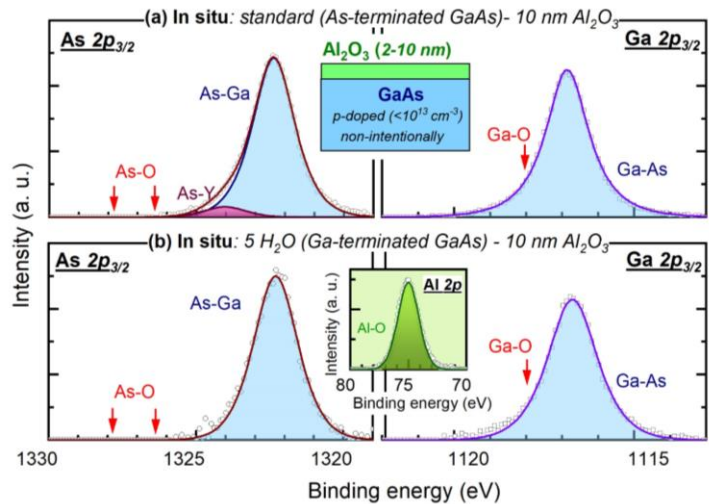


Fig.1. Photoelectron spectra for the As 2*p*_{3/2} (left) and Ga 2*p*_{3/2} (right) core-level features after *in situ* deposition of 10 nm Al₂O₃ on GaAs surface at 300 °C. The Al₂O₃ deposition begins with 1 TMA pulse (standard) (a) and 5 H₂O pulses (b). Inset: (a) Al₂O₃/GaAs structure; (b) Photoelectron spectra for the Al 2*p*_{5/2}.